



# 0D Bi nanodots/2D Bi<sub>3</sub>NbO<sub>7</sub> nanosheets heterojunctions for efficient visible light photocatalytic degradation of antibiotics: Enhanced molecular oxygen activation and mechanism insight

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## ABSTRACT

Solar-driven molecular oxygen activation is a promising and low energy-cost way for environmental remediation. To fabricate semimetal/semiconductor composite photocatalyst is an effective strategy to accelerate the transfer and separation of photogenerated carriers for boosting molecular oxygen activation. Herein, we report a 0D Bi nanodots/2D Bi<sub>3</sub>NbO<sub>7</sub> nanosheets heterostructured composite with enhanced molecular oxygen activation under visible light irradiation, which was synthesized by a two-step wet chemical method. Transmission electron microscopy (TEM) analysis shows that the Bi nanodots with diameters of 2–5 nm were uniformly distributed on the surface of Bi<sub>3</sub>NbO<sub>7</sub> nanosheets. More importantly, both experiments and density functional theory (DFT) calculations confirm that a strong covalent interaction existed between the Bi atom of Bi nanodots and Bi–O layer on the surface of the Bi<sub>3</sub>NbO<sub>7</sub> nanosheets, which enhanced the visible light absorbability of the composite, fostered the transfer and separation of its interfacial photogenerated carriers, and promoted the activation of molecular oxygen into superoxide radicals ( $\cdot\text{O}_2^-$ ) and singlet oxygen ( $^1\text{O}_2$ ) by the composite under visible light illumination for degradation of ciprofloxacin (CIP). The photocatalytic degradation rate of CIP by the Bi/Bi<sub>3</sub>NbO<sub>7</sub> composites is 4.58 times higher than that by the pristine Bi<sub>3</sub>NbO<sub>7</sub>. The Bi/Bi<sub>3</sub>NbO<sub>7</sub> photocatalyst still revealed high photocatalytic activity even after five cycles. This work elucidates the mechanism of molecular oxygen activation over 0D/2D semimetal-semiconductor system and provides a promising approach for designing high efficient 0D/2D photocatalysts toward sustainable environmental remediation.

## 1. Introduction

In the past decades, the abuse of antibiotics causes great concerns on natural water bodies and human health. Nowadays, semiconductor photocatalysis has been regarded as a promising approach to decompose the excess antibiotics in environmental water bodies [1–3]. It is well known that activation of molecular oxygen process acts an important role in photocatalytic degradation of antibiotics. The effective capture of molecular oxygen is capable of suppressing the recombination of photo-induced electron-hole pairs and generates reactive oxygen species (ROS), such as superoxide ( $\cdot\text{O}_2^-$ ), singlet oxygen ( $^1\text{O}_2$ ), hydroxyl ( $\cdot\text{OH}$ ), which are crucial active sources of photocatalytic antibiotics degradation [4–6]. Thus, the construction of novel visible-light responsive photocatalysts based molecular oxygen activation for effectively antibiotics degradation still remains a huge challenge.

Bismuth-based photocatalysts have triggered widespread concerns

for their high charge separation efficiency and efficient utilization of solar spectrum [7–11]. Typically, on account of the unique layered structure, suitable band gap (2.5–2.75 eV) and high photochemical stability, bismuth niobates (Bi<sub>3</sub>NbO<sub>7</sub>) has been regarded as a promising semiconductor photocatalyst for environmental pollutants removal and antibiotics degradation [12–14]. Our group and Hu et al. had reported Bi<sub>3</sub>NbO<sub>7</sub> nanoparticles and porous Bi<sub>3</sub>NbO<sub>7</sub> nanosheets, which showed good adsorption ability and visible light photocatalytic performance [15,16]. However, the photocatalytic performance of Bi<sub>3</sub>NbO<sub>7</sub> is markedly restricted by the insufficient of solar spectrum absorption and poor molecular oxygen activation efficiency. Element doping and heterojunctions construction are effective strategies to improve the photocatalytic activity of this semiconductor photocatalyst system [17,18]. In addition, low dimension metal nanoparticles deposition on semiconductor nanostructure is a meaningful method to extend the visible light responsive range and the molecular oxygen activation efficiency of

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$\text{Bi}_3\text{NbO}_7$ .

Noble metal nanoparticles such as Ag, Au and Pt modified semiconductors have been widely applied to accelerate the solar spectrum absorption and photocatalytic activity of photocatalysts. The semimetal Bi, as a direct plasmonic photocatalyst, has been confirmed to exhibit stable photocatalytic performance due to the SPR property, similar to many noble metal nanoparticles [19–23]. However, the photocatalytic efficiency of single semimetal Bi photocatalyst calls for further improvements in photochemical field. The zero-dimension Bi nanodots deposited on the surface of semiconductor photocatalysts could boost the interfacial charge flow by constructing Schottky barriers at the interface of the metal/catalyst systems and play as an electron trap to separate photoinduced carriers [24–27]. Recently, deposition of Bi nanoparticles onto semiconductor photocatalysts, such as  $\text{TiO}_2$ , BiOI, BiOCl,  $\text{Bi}_2\text{WO}_6$ , and CdS displayed high photocatalytic activities for the environmental pollutants removal and hydrogen evolution [28–35]. It might be a fantasy approach to boost the photocatalytic performance and molecular oxygen activation of the  $\text{Bi}_3\text{NbO}_7$  by constructing 0D Bi nanodots/2D  $\text{Bi}_3\text{NbO}_7$  nanosheets heterojunction system.

In the present report, we successfully fabricated 0D/2D Bi/ $\text{Bi}_3\text{NbO}_7$  heterojunction photocatalysts, which displayed superior molecular oxygen activation ability for photocatalytic antibiotics degradation. Subsequently, the semimetal Bi/ $\text{Bi}_3\text{NbO}_7$  homogeneous junction and charge difference distribution between  $\text{Bi}_3\text{NbO}_7$  and Bi were calculated by theoretical calculations and the 0D Bi nanodots decorated  $\text{Bi}_3\text{NbO}_7$  nanosheets exhibited remarkably enhanced photocatalytic performance for ciprofloxacin (CIP) degradation. Furthermore, PL, time-resolved transient PL decay spectra, and photo-electrochemistry analysis were used to analyze the interfacial charge separation of Bi/ $\text{Bi}_3\text{NbO}_7$ . Consequently, this work could provide enlightenment for synthesizing 0D/2D Bi-based photocatalysts with high photocatalytic performance.

## 2. Experimental

### 2.1. Synthesis

$\text{Bi}_3\text{NbO}_7$  nanosheets were synthesized by a facile hydrothermal method. 1.455 g of  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  was dissolved into 10 mL glacial acetic acid as solution A and 0.538 g of  $\text{C}_{10}\text{H}_5\text{NbO}_{20}$  was added into 40 mL methanol as solution B. Subsequently, solution A was slowly added into as solution B under vigorous stirring followed by 30 mL benzyl alcohol after 30 min. The pH value of suspension was adjusted to 13.2 using KOH solution. The suspension was transferred to PPL-lined autoclave and heated at 180 °C for 24 h. The yellow products were washed with ethanol for five times and dried at 80 °C for 4 h. Bi/ $\text{Bi}_3\text{NbO}_7$  nanocomposites were fabricated by a facile solvothermal method. (0.005, 0.02, 0.035, 0.05) mmol  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  was dissolved in 65 mL ethylene glycol (EG) for 30 min. Then 0.42 g  $\text{Bi}_3\text{NbO}_7$  powder was added into the above solution under ultrasonic dispersion for 2 h and heated at 180 °C for 8 h. The obtained composites were washed with ethanol for five times and dried at 80 °C for 12 h. The contents of Bi in Bi/ $\text{Bi}_3\text{NbO}_7$  composites were 1, 4, 7, and 10% by mass ratio.

### 2.2. Characterization

The morphologies and high angle angular dark field-scanning transmission electron microscopy (HAADF-STEM) images of the sample were examined by transmission electron microscopy (Talos F200S, Thermo Fisher). The specific surface areas of the samples were tested by Micromeritics ASAP 2020. The absorption edges of samples were tested by a UV–vis spectrophotometer (Lambda 750S). The crystallinity of samples was analyzed by X-ray diffraction (XRD) on a X'Pert PRO diffractometer with Cu K $\alpha$  radiation. X-ray photoelectron spectra (Thermo Fisher Multi-lab 2000) were tested on a monochromatic Al K $\alpha$  source and the C(1s) binding energy is 284.6 eV. Photoluminescence spectra were recorded by fluorescence spectrophotometer (Shimadzu RF-5301)

and the Time-resolved transient PL decay was tested by Edinburgh FLSP920 with an excitation wavelength of 312 nm. The EPR (electron paramagnetic resonance) spectra were recorded on a Bruker A-300 spectrometer. Electrochemical impedance spectroscopy and transient photocurrent were analyzed by an electrochemical workstation (chi660E) in the electrolyte was 0.5 M  $\text{Na}_2\text{SO}_4$ .

### 2.3. Photocatalytic activity and molecular oxygen activation

The photocatalytic performance of the Bi/ $\text{Bi}_3\text{NbO}_7$  catalysts were measured by the degradation of CIP, using 300 W Xe lamp as light source with 400 nm cutoff filter. Typically, 50 mg photocatalysts was placed in 100 mL CIP solution (10 mg/L). The concentration of CIP in aqueous solution was analyzed by UV–vis spectrophotometer (Thermo Orion AquaMate 8000) at wavelength of 277 nm. Total organic carbon (TOC) was analysed by total organic carbon analyzer (Elementar, Germany). Determination of the intermediates of CIP was carried out on a LC–MS system (Agilent Quadrupole 6120).

The ESR experiments were tested by electron paramagnetic resonance spectrometer (Bruker, A-300). Typically, 20 mg catalysts were dispersed in 1 mL of water or methanol, and 45  $\mu\text{L}$  of 5,5-Dimethyl-1-pyrroline-N-oxide (DMPO) was added before visible light irradiation. 2,2,6,6-tetramethylpiperidine (TEMP) was used to detect the singlet oxygen ( $^1\text{O}_2$ ). The 20  $\mu\text{M}$  of nitroblue tetrazolium (NBT) was applied to quantize the amount of  $\cdot\text{O}_2^-$ , which was quantitatively analyzed by measuring the dissipation of NBT on a UV–vis spectrophotometer at 259 nm.

### 2.4. Computational details

Vienna ab initio simulation package (VASP) based on density functional theory (DFT) was used to calculate the electronic structure of  $\text{Bi}_3\text{NbO}_7$  and Bi/ $\text{Bi}_3\text{NbO}_7$ . The exchange-correction function was measured by the Perdew-Burke-Ernzerhof (PBE) formulation of the generalized gradient approximation (GGA). The cutoff energy of atomic wave functions was 400 eV and the k-meshes  $3 \times 2 \times 2$  and  $3 \times 2 \times 4$  are performed in the calculations of  $\text{Bi}_3\text{NbO}_7$  and Bi/ $\text{Bi}_3\text{NbO}_7$ , respectively. During the optimizations, the energy and force converged to  $10^{-5}$  eV/atom and 0.01 eV/Å, respectively. Then 8 Bi atoms were supported on the surface of  $\text{Bi}_3\text{NbO}_7$ .

## 3. Results and discussion

As shown in Fig. 1, the diffraction peaks of the  $\text{Bi}_3\text{NbO}_7$  at 2 $\theta$  degree of 28.19°, 32.66°, 46.86°, 55.59° have been observed, which matched

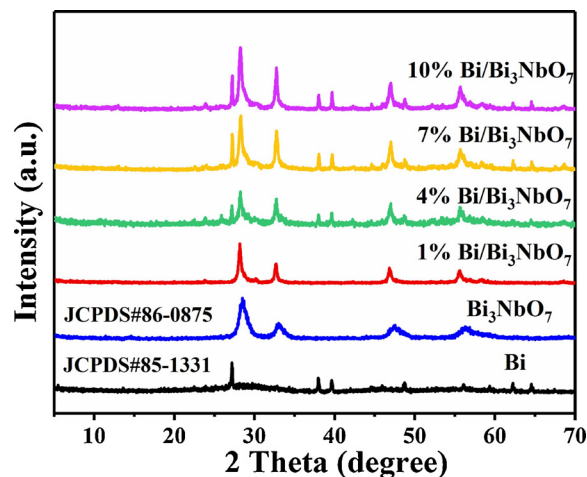


Fig. 1. XRD patterns of the pure  $\text{Bi}_3\text{NbO}_7$ , Bi and X% Bi/ $\text{Bi}_3\text{NbO}_7$  (X = 1, 4, 7, 10).

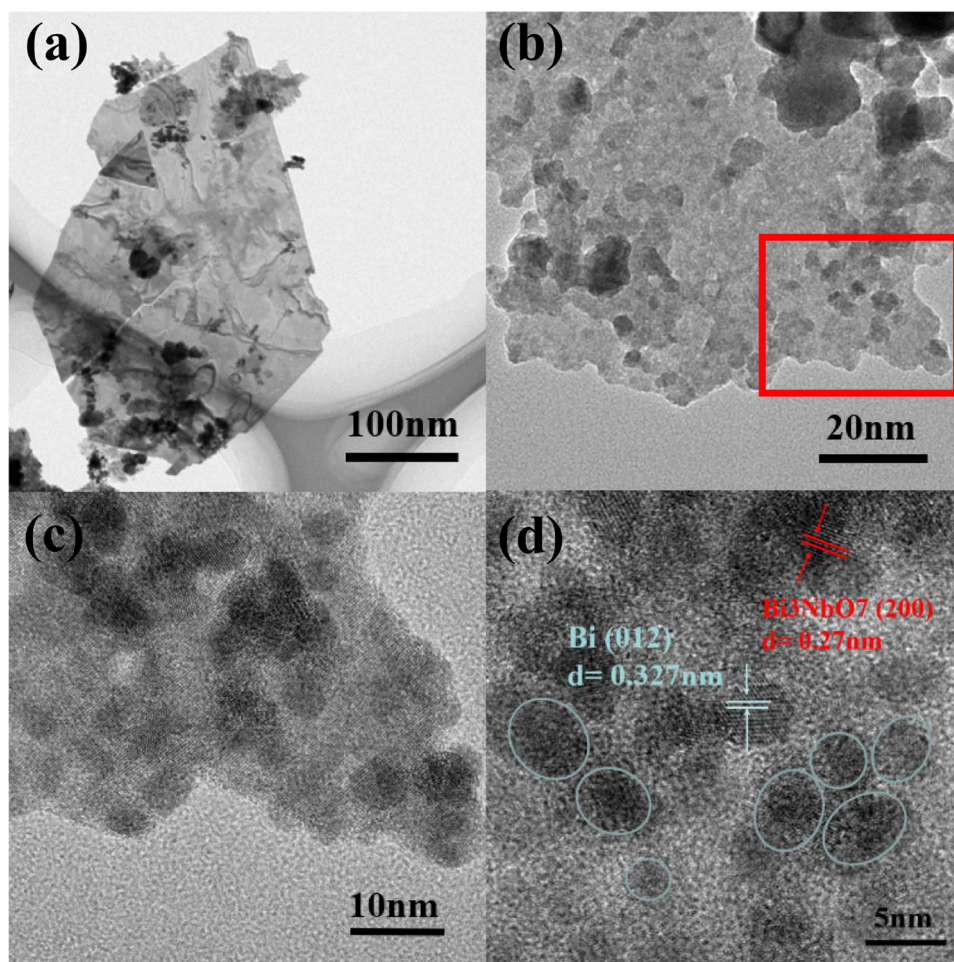


Fig. 2. TEM images of (a)  $\text{Bi}_3\text{NbO}_7$ , (b and c) 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$ . HRTEM image of (d) 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$ .

well with the (111), (200), (220), (311) planes of the cubic  $\text{Bi}_3\text{NbO}_7$  (JCPDS, No. 86-0875) [16]. For the X%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  ( $X = 4, 7$ , and 10) composites, all the peaks of  $\text{Bi}_3\text{NbO}_7$  were clearly observed at  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites, which forecast that the introduction of Bi has no influence to the crystalline structure of  $\text{Bi}_3\text{NbO}_7$ . Meanwhile, the minor peaks at  $27.27^\circ$ ,  $38.13^\circ$ ,  $39.74^\circ$ ,  $48.86^\circ$  were founded, which can be assigned to (012), (104), (110), (202) planes of metallic Bi (JCPDS, No. 85-1331) [33]. Remarkably, no obvious diffraction peaks for Bi could be observed in the pattern of 1%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$ , which might be resulted from the infinitesimal Bi nanoparticles in 1%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  samples.

The morphologies of the  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites were investigated by SEM and TEM. Fig. S1 can lead to the fact that the pure  $\text{Bi}_3\text{NbO}_7$  displays 2D structure [16]. More detailed informations of the  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  samples could be observed from TEM images. From Fig. 2a, the TEM image of  $\text{Bi}_3\text{NbO}_7$  is in good agreement with the sheet-like structure in the SEM image and the thickness is quite thin. Fig. 2b and c indicates that Bi nanodots are well dispersed on the surface of  $\text{Bi}_3\text{NbO}_7$  nanosheet with diameters is 2–5 nm. The uniformly loaded Bi nanodots were beneficial to the interfacial charge flow from  $\text{Bi}_3\text{NbO}_7$  nanosheet to Bi nanodots. It can be clear observed in Fig. 2d that the lattice fringes with spacing of 0.326 nm is consistent with the (012) crystal planes of semimetal Bi (JCPDS, No. 85-1331) [31]. HAADF-STEM image and corresponding EDX mappings of 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  was used to inspect the elemental composition and distribution (Fig. S2). The mapping images display a homogeneous distribution of Bi, Nb, and O elements in 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$ .

The FTIR spectra of the  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites were exhibited in Fig. 3. As shown in the spectrum of  $\text{Bi}_3\text{NbO}_7$ , the broad

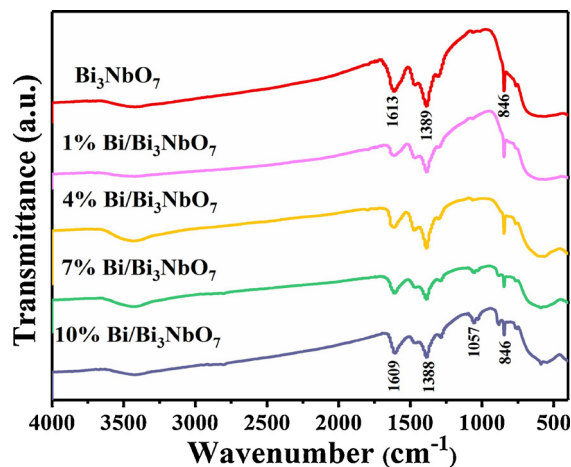


Fig. 3. FT-IR spectra of the  $\text{Bi}_3\text{NbO}_7$  and X%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  ( $X = 1, 4, 7, 10$ ).

bands ranges from 600–1200  $\text{cm}^{-1}$  are attributes to Nb–O, Nb–O–Nb and Bi–O stretching vibrations and the band at 1613  $\text{cm}^{-1}$  could correspond to the water molecules adsorbed on the surface of the  $\text{Bi}_3\text{NbO}_7$  [31]. The weakened peak at 1388  $\text{cm}^{-1}$  and a new peak appeared at 1057  $\text{cm}^{-1}$  demonstrates the presence of metallic Bi on the  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites. The FTIR results declare that the structure of  $\text{Bi}_3\text{NbO}_7$  does not change with the introduction of semimetal Bi, which is in consensus with the XRD analysis.

XPS was further measured to investigate the surface chemical states



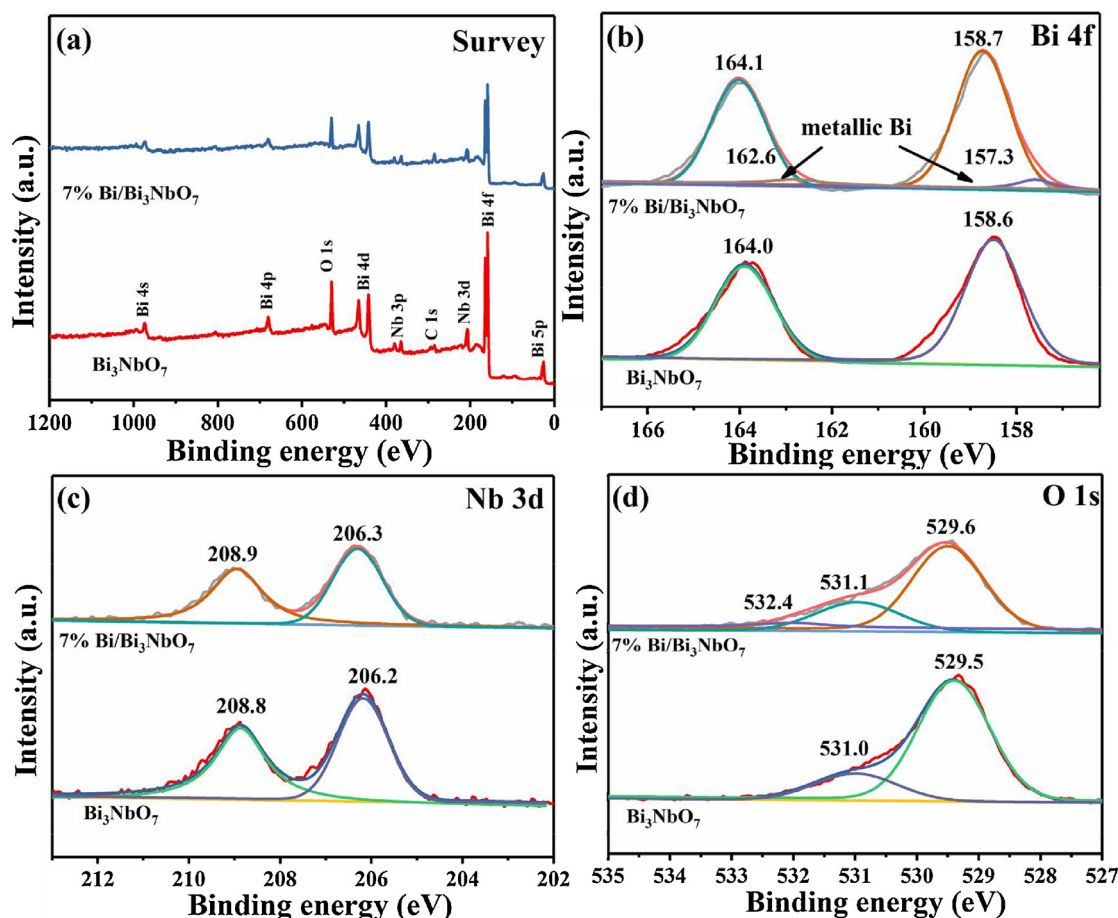


Fig. 4. XPS spectra of the  $\text{Bi}_3\text{NbO}_7$  and 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composites. (a) the survey scan, (b) Bi 4f, (c) Nb 3d, and (d) O 1s.

of  $\text{Bi/Bi}_3\text{NbO}_7$  composite. As shown in Fig. 4a, the survey spectrum illustrates that composition of the  $\text{Bi}_3\text{NbO}_7$  and 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composites included Bi, Nb and O. Fig. 4b displays Bi 4f spectra in the 7%  $\text{Bi/Bi}_3\text{NbO}_7$  and  $\text{Bi}_3\text{NbO}_7$  sample. It can be observed that two peaks at 164.0 and 158.6 eV are referred to the binding energies of Bi  $4f_{5/2}$  and Bi  $4f_{7/2}$ , respectively, which are the characteristics of  $\text{Bi}^{3+}$  in  $\text{Bi}_3\text{NbO}_7$ . The Bi 4f peaks in 7%  $\text{Bi/Bi}_3\text{NbO}_7$  can be divided into four peaks at 158.7, 164.1 eV, and 157.3, 162.6 eV, respectively. Those peaks located at 157.3 and 162.6 eV in the XPS spectrum of 7%  $\text{Bi/Bi}_3\text{NbO}_7$  can be attributed to semimetal Bi nanodots on the surface of  $\text{Bi}_3\text{NbO}_7$  nanosheets, which is consistent with the results of HRTEM analysis (Fig. 2d) [24]. The binding energy at 206.2 and 208.8 eV are assigned to Nb  $3d_{3/2}$  and Nb  $3d_{5/2}$  peaks, respectively (Fig. 4c), which are well matched with the  $\text{Nb}^{5+}$  in  $\text{Bi}_3\text{NbO}_7$  and 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composites [16]. When semimetal Bi was introduced, the two peaks of  $\text{Nb}^{5+}$  shifted to 208.9 and 206.3 eV in 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composites, respectively. As shown in Fig. 4d, the O 1s spectrum of  $\text{Bi}_3\text{NbO}_7$  can be divided into two peaks and the peak located at 529.5, and 531.0 eV could be assigned to Bi–O bond and Nb–O bond of  $\text{Bi}_3\text{NbO}_7$  crystal structures, respectively [31]. Interestingly, a new peak at around 532.4 eV in the O 1s spectrum is observed for the 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composites, which can be ascribed to oxygen in hydrated species OH on the surface [33]. Meanwhile, the XPS of  $\text{Bi}_3\text{NbO}_7$  composites before and after solvothermal reaction were also analyzed. As shown in Fig. S3a–c, the high-resolution spectra of Bi 4f, Nb 3d and O 1s changed little and overlapped basically, which indicated that the second solvothermal process by adding ethylene glycol as reducing agent had little influence on the physicochemical property of  $\text{Bi}_3\text{NbO}_7$ . Furthermore, the EPR spectra of  $\text{Bi}_3\text{NbO}_7$  before and after solvothermal reaction in Fig. S3d exhibited that there is no obvious peak of oxygen vacancy after solvothermal reaction. These results

illustrated the existence of OD Bi nanodots reduced by  $\text{Bi}^{3+}$  of adding Bi ( $\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  in second solvothermal process, further indicating the formation of  $\text{Bi/Bi}_3\text{NbO}_7$  heterostructure.

UV-vis absorption spectra of pristine  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi/Bi}_3\text{NbO}_7$  composites were exhibited in Fig. 5. From Fig. 5a, the absorption edge of pristine  $\text{Bi}_3\text{NbO}_7$  is approximately located at 462 nm. The absorption edges of the  $\text{Bi/Bi}_3\text{NbO}_7$  composites exhibited a prominent redshift, which confirmed that the  $\text{Bi/Bi}_3\text{NbO}_7$  composites further enhanced the visible light absorption [34]. As shown inset of Fig. 5a, the optical absorption of the obtained samples changed with increasing the Bi content in the samples, which is consistent with their color changes from light yellow to grey. Notably, a typical SPR absorption region of semimetal Bi at about 500 nm can be observed in Fig. 5a. According to the previous literature, the enhanced visible light absorption ranges from 450 to 600 nm can be easily detected from the SPR effects of semimetal Bi [26]. Instead, the  $\text{Bi/Bi}_3\text{NbO}_7$  composites almost absorb full region of light, exhibits especially enhanced absorption in the visible light range (450–700 nm), which can be directly ascribed to SPR absorption from OD Bi nanodots [23]. Fig. 5b indicates that the band gap ( $E_g$ ) of the sample is located at 2.69 eV for  $\text{Bi}_3\text{NbO}_7$ . In general, the OD Bi nanodots deposited on  $\text{Bi}_3\text{NbO}_7$  can enhance the visible light absorption and molecular oxygen activation of pristine  $\text{Bi}_3\text{NbO}_7$ .

Additionally, density functional theory (DFT) was applied to explore the electron properties of  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi/Bi}_3\text{NbO}_7$  composites theoretically. The cell models of  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi/Bi}_3\text{NbO}_7$  were shown in Fig. 6a, respectively. As shown in Fig. 6b, the upper valence band is the hybridization of Bi 6s, Nb 4d and O 2p orbitals, respectively [36–38] and the conduction band from 2.6 to 10.9 eV is mostly contributed by the hybridization of Bi 5d, Nb 4d and O 2p orbital. The calculated band gap value of  $\text{Bi}_3\text{NbO}_7$  (2.56 eV) tends to be smaller than the

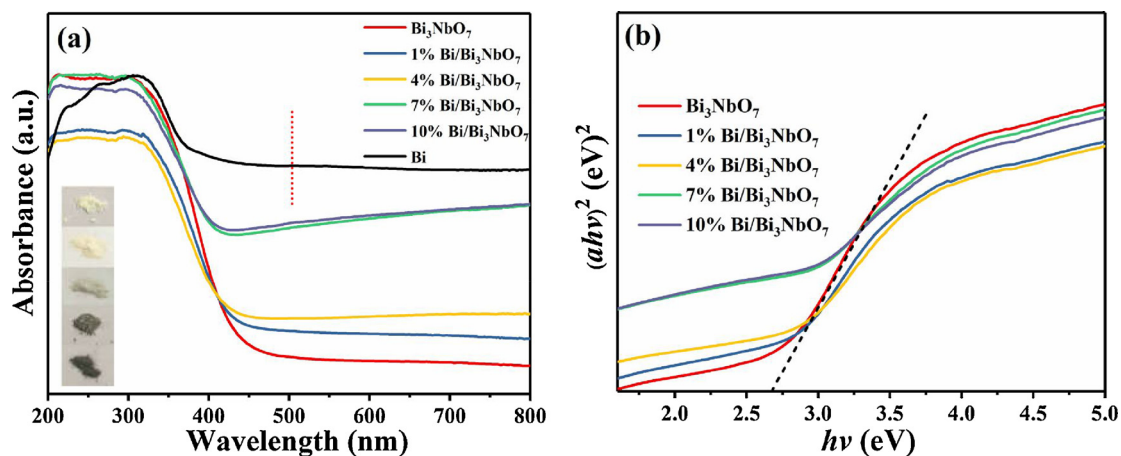


Fig. 5. (a) UV-vis diffuse reflectance spectra and (b) Plots of  $(\alpha h\nu)^2$  versus  $h\nu$  for of the  $\text{Bi}_3\text{NbO}_7$  and serial  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites.

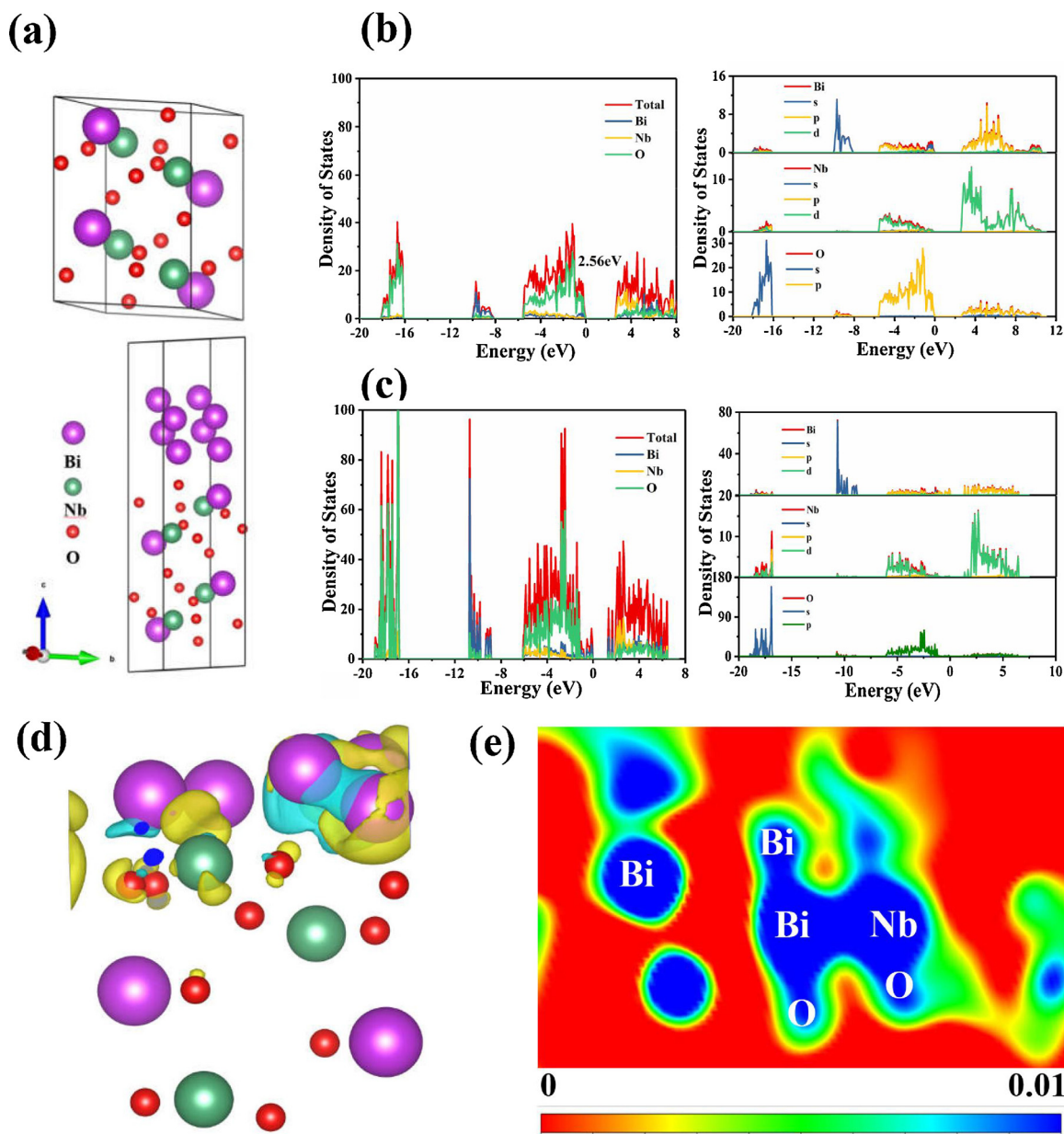


Fig. 6. (a) Optimized geometric structures of  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  nanocomposite. The TDOS and corresponding PDOS of (b)  $\text{Bi}_3\text{NbO}_7$  and (c)  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  nanocomposite. (d) Charge difference distribution and (e) electronic location function (ELF) of  $\text{Bi}/\text{Bi}_3\text{NbO}_7$ .

experimental value (2.69 eV), owing to the restriction of the DFT calculation. Meanwhile, we calculate the density of state (DOS) of Bi/Bi<sub>3</sub>NbO<sub>7</sub> [39]. In comparison with the DOS for perfect Bi<sub>3</sub>NbO<sub>7</sub>, the density of states at valence band maximum (VBM) of Bi/Bi<sub>3</sub>NbO<sub>7</sub> was significantly increased by the contribution of the orbital from Bi element. The conduction band of Bi/Bi<sub>3</sub>NbO<sub>7</sub> shifted to lower energy levels, which is beneficial to promote the generation of photogenerated charge carriers [26,40]. Fig. 6d displays the charge difference distribution of Bi atom layer on Bi<sub>3</sub>NbO<sub>7</sub> unit cell. It's obvious that the Bi–O layer of Bi<sub>3</sub>NbO<sub>7</sub> donates abundant free electrons to metal Bi atom. From Fig. 6e, the electronic location function (ELF) demonstrates the existence of a strong covalent interaction between the Bi atom of Bi nanodots and Bi–O layer on the surface of the Bi<sub>3</sub>NbO<sub>7</sub> nanosheets, forming a charge transfer channel to transfer from Bi<sub>3</sub>NbO<sub>7</sub> to metal Bi nanodots [40]. To confirm the effect of Bi nanodots on the molecular oxygen activation, the charge distributions of Bi<sub>3</sub>NbO<sub>7</sub> and Bi/Bi<sub>3</sub>NbO<sub>7</sub> with O<sub>2</sub> adsorption were also calculated as presented in Fig. S4. When Bi atom is introduced, the charge distribution between the O<sub>2</sub> molecules and Bi layer displays a typical funnel-like three-dimensional structure, which indicated the outermost electrons tend to accumulate on oxygen molecules adsorbed on the surface of Bi layer of Bi/Bi<sub>3</sub>NbO<sub>7</sub>, which attributes to the activation of O<sub>2</sub> into •O<sub>2</sub><sup>−</sup>.

The photocatalytic activities of Bi<sub>3</sub>NbO<sub>7</sub> and serial Bi/Bi<sub>3</sub>NbO<sub>7</sub> composites were evaluated by the degradation of typical antibiotics ciprofloxacin (CIP). Only 4% CIP in the solution were degraded within 120 min under visible light irradiation in the absence of catalyst. As shown in Fig. 7a, it is obvious that 36, 51, 64, 86, and 74% of CIP were

decomposed by Bi<sub>3</sub>NbO<sub>7</sub> and serial Bi/Bi<sub>3</sub>NbO<sub>7</sub> composites with different content of Bi nanodots, respectively. Notably, the 7% Bi/Bi<sub>3</sub>NbO<sub>7</sub> composite exhibited the highest photocatalytic performance. From Fig. S5, the peak at 277 nm (maximum absorption wavelength of CIP) was rapidly weakened, reflecting that the CIP molecular was decomposed by the reactive oxygen species generated by Bi/Bi<sub>3</sub>NbO<sub>7</sub> photocatalysts. The corresponding degradation rate constants (min<sup>−1</sup>) of CIP were also presented in Fig. 7b, and obviously, the 7% Bi/Bi<sub>3</sub>NbO<sub>7</sub> exhibited the highest rate (0.01427) in CIP degradation progress, which is about 4.58 times higher than that of pure Bi<sub>3</sub>NbO<sub>7</sub>. TOC removal was chosen as a mineralization index to analyse the CIP degradation progress. As shown in Fig. 7c, the 7% Bi/Bi<sub>3</sub>NbO<sub>7</sub> samples get a total TOC removal efficiency of 53% after 180 min, which is higher than that of pure Bi<sub>3</sub>NbO<sub>7</sub> (19%) under same conditions, indicating that the Bi/Bi<sub>3</sub>NbO<sub>7</sub> composites present enhanced mineralization ability in antibiotics degradation process. For comparison, photocatalytic CIP degradation with commercial TiO<sub>2</sub> (P25), pure Bi<sub>3</sub>NbO<sub>7</sub>, and Bi was also measured with the same conditions under the visible light irradiation, respectively. As shown in Fig. 7d, the 7% Bi/Bi<sub>3</sub>NbO<sub>7</sub> composite exhibited the higher-up photocatalytic performance over pure Bi<sub>3</sub>NbO<sub>7</sub>, semimetal Bi, and P25, confirming that constructing strong coupled 0D/2D structure significantly improved the photocatalytic activities.

In this study, the degradation intermediates of CIP during the photocatalytic process were detected by LC–MS. LC–MS chromatogram and *m/z* of degraded products are listed in Fig. S6. By referencing relevant reports, the possible photocatalytic degradation pathways of CIP are displayed in Scheme 1. Three major degradation pathways are

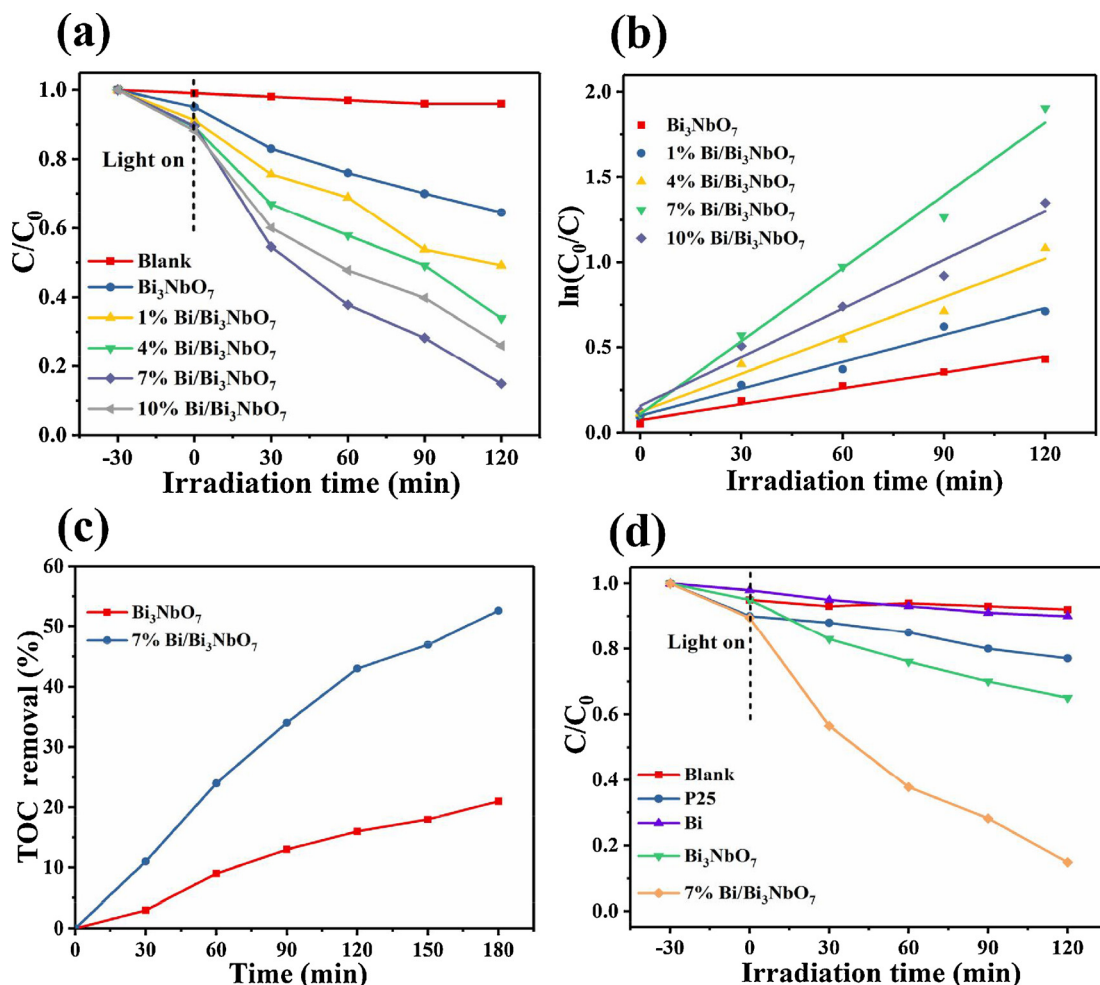
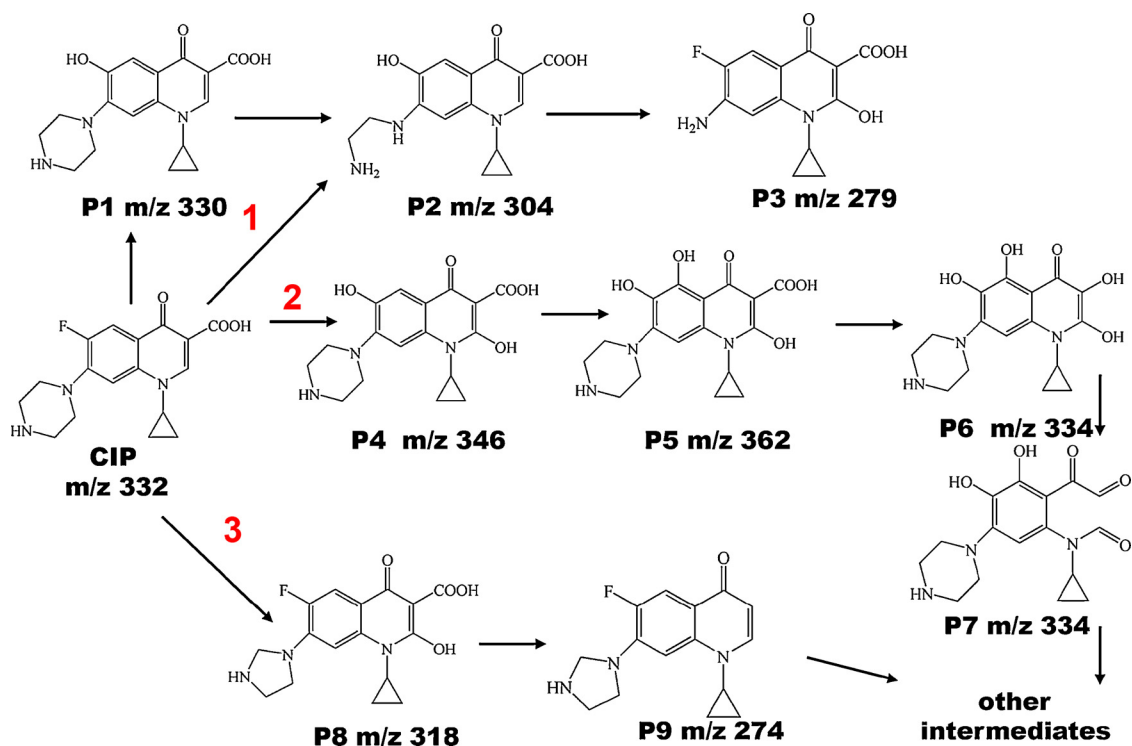


Fig. 7. (a) Photocatalytic degradation curves of CIP solution and (b) first-order rate constants of CIP degradation by series of Bi/Bi<sub>3</sub>NbO<sub>7</sub> composites. (c) TOC removal curves of Bi<sub>3</sub>NbO<sub>7</sub> and 7% Bi/Bi<sub>3</sub>NbO<sub>7</sub> photocatalysts and (d) photocatalytic degradation curves of CIP by different catalysts.





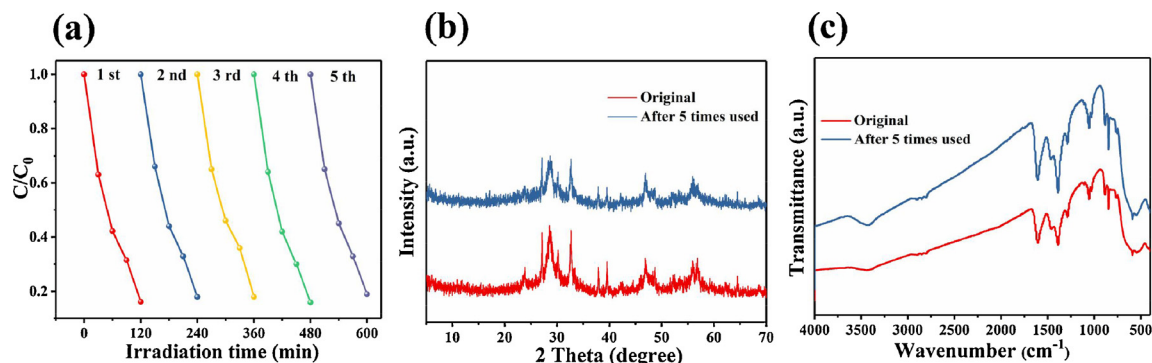
Scheme 1. Possible photocatalytic degradation pathways of CIP.

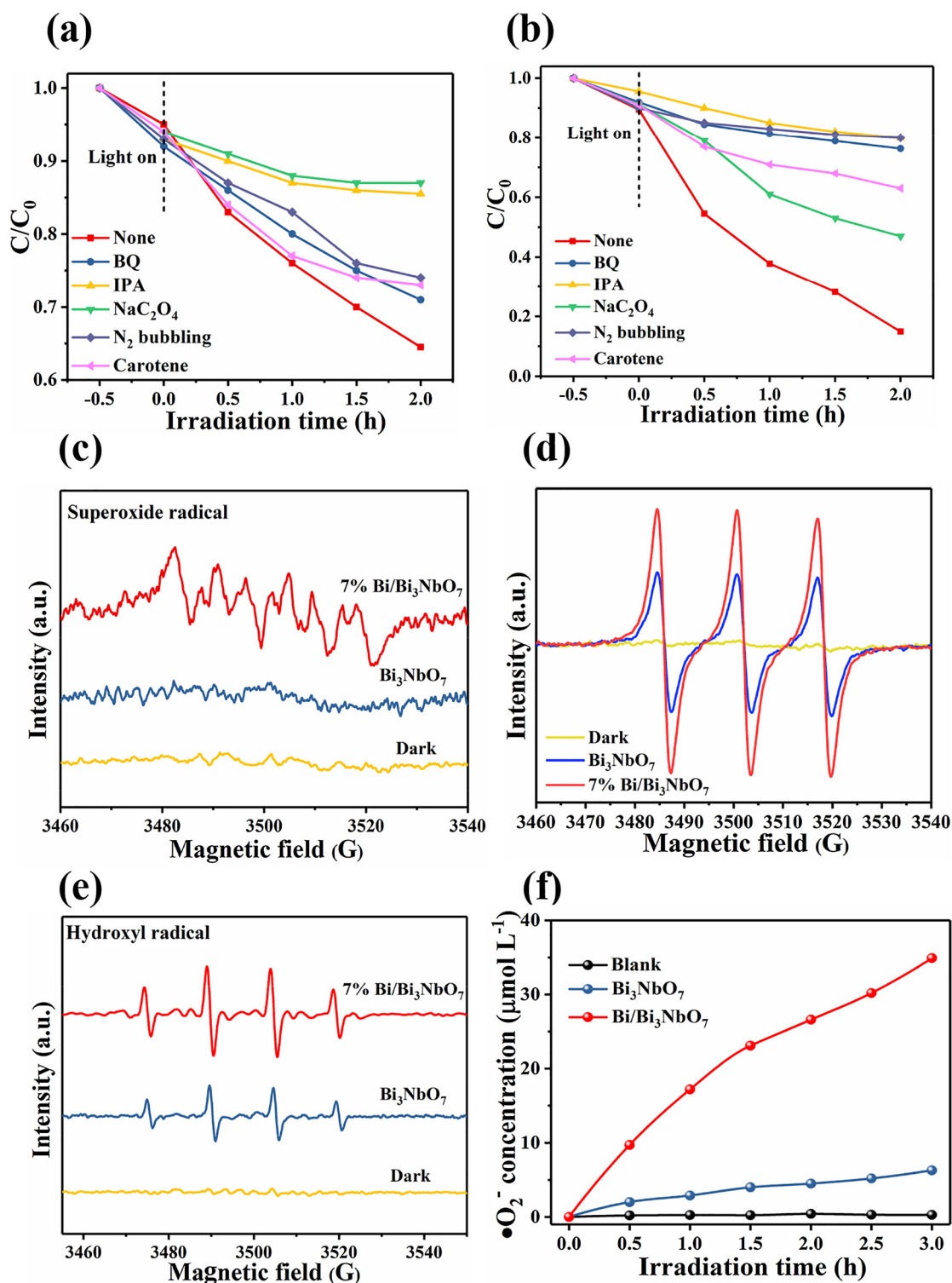
presented. Pathway 1 : CIP  $\rightarrow$  P1 ( $m/z = 330$ )  $\rightarrow$  P2 ( $m/z = 304$ )  $\rightarrow$  P3 ( $m/z = 279$ ). In this degradation process, the cleavage of the piperazine ring mainly took place. Pathway 2 : CIP  $\rightarrow$  P4 ( $m/z = 346$ )  $\rightarrow$  P5 ( $m/z = 362$ )  $\rightarrow$  P6 ( $m/z = 334$ )  $\rightarrow$  P7 ( $m/z = 334$ ). This pathway was mainly a hydroxylation process. CIP was subjected to varying degrees of hydrolysis and generated P4, P5, and P6. Then the quinolone ring was broken, resulting in the formation of P7. Pathway 2 : CIP  $\rightarrow$  P4 ( $m/z = 346$ )  $\rightarrow$  P5 ( $m/z = 362$ )  $\rightarrow$  P6 ( $m/z = 334$ )  $\rightarrow$  P7 ( $m/z = 334$ ). Pathway 3 : CIP  $\rightarrow$  P8 ( $m/z = 318$ )  $\rightarrow$  P9 ( $m/z = 274$ ). The formation of P8 ( $m/z = 318$ ) with a five-membered ring structure in piperazine ring moiety during CIP degradation was proposed. Comparing with P8, the molecular weight of intermediate P9 was decreased by 44, just corresponding to one molecule of  $\text{CO}_2$  [41–43]. Both pathways would finally mineralize CIP into  $\text{CO}_2$  and  $\text{H}_2\text{O}$ , as well as inorganic oxidized N and F [41].

The photocatalytic stability of  $\text{Bi/Bi}_3\text{NbO}_7$  composites was also evaluated by the degradation of CIP solution (Fig. 8). The 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composite was still maintained at 82% after five cycles, which indicates that the  $\text{Bi/Bi}_3\text{NbO}_7$  composites are a stable catalyst for the photocatalytic antibiotics degradation. Moreover, the XRD and FT-IR patterns of the 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composite before and after

photocatalytic experiments were also analyzed as shown in Fig. 8b and c. It was obvious that the XRD and FT-IR patterns of the 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composite after five times had almost no prominent decrease compared with the fresh one.

Radical species trapping experiments were measured to detect the generated reactive oxygen species in photocatalytic CIP degradation. Benzoquinone, Carotene, Sodium oxalate and Isopropanol were used as  $\cdot\text{O}_2^-$ ,  $^1\text{O}_2$ ,  $\text{h}^+$  and  $\cdot\text{OH}$  scavengers, respectively [4]. From Fig. 9a, the CIP degradation rates of  $\text{Bi}_3\text{NbO}_7$  were dropped to 29%, 27%, 13%, 15%, 25%, and 36% in the presence of Benzoquinone, Carotene, Sodium oxalate, Isopropanol,  $\text{N}_2$  bubbling and without scavengers respectively, confirming that  $\cdot\text{O}_2^-$  and  $^1\text{O}_2$  plays no essential contribution in CIP degradation. However, the CIP degradation rates of 7%  $\text{Bi/Bi}_3\text{NbO}_7$  degraded slowly in the presence of Benzoquinone, Carotene,  $\text{N}_2$  bubbling and Isopropanol, confirming that  $\cdot\text{O}_2^-$  and  $^1\text{O}_2$  together acted the important role in the antibiotics degradation (Fig. 9b) and  $\cdot\text{OH}$  contributes some in degradation process. Furthermore, electron spin resonance (ESR) were used to further investigate the existence of reactive oxygen species [32]. Fig. 9c shows the ESR signals of DMPO- $\cdot\text{O}_2^-$  for  $\text{Bi}_3\text{NbO}_7$  in dark and  $\text{Bi}_3\text{NbO}_7$ , 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composite under visible light irradiation. It is obvious that the 7%  $\text{Bi/Bi}_3\text{NbO}_7$

Fig. 8. (a) Photocatalytic recycle experiments for the 7%  $\text{Bi/Bi}_3\text{NbO}_7$  sample. XRD patterns (b) and FTIR spectra (c) of original and used 7%  $\text{Bi/Bi}_3\text{NbO}_7$  composites.

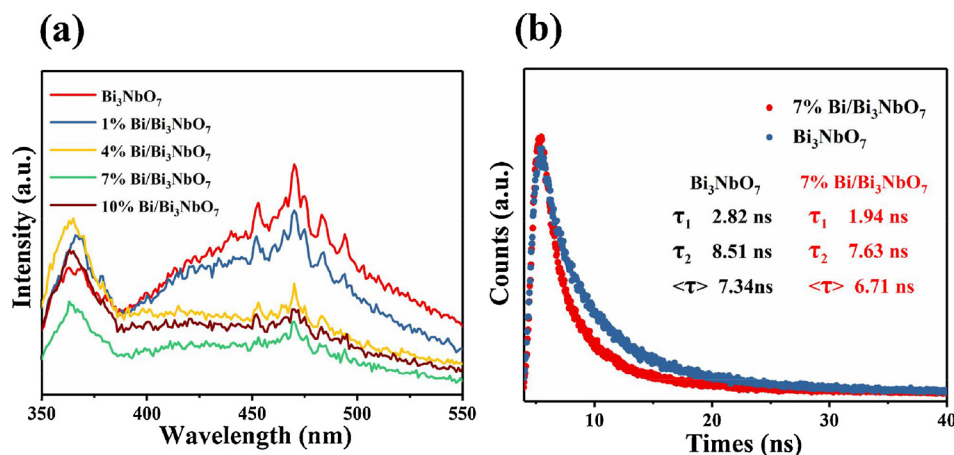


**Fig. 9.** Photocatalytic activities of (a)  $\text{Bi}_3\text{NbO}_7$  and (b) 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites for the CIP degradation with disparate scavengers; ESR spectra of the (c)  $\text{DMPO} \cdot \text{O}_2^-$ , (d)  $\text{DMPO} \cdot {}^1\text{O}_2$  and (e)  $\text{DMPO} \cdot \text{OH}$  for  $\text{Bi}_3\text{NbO}_7$  and the 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composite under visible light irradiation or in dark. (f) Quantitative determination of  $\text{O}_2^-$ 's generation for different samples.

composite displayed the powerful intensities as compared to  $\text{Bi}_3\text{NbO}_7$ . Meanwhile, singlet oxygen ( ${}^1\text{O}_2$ ) was also detected by using TEMP as the trapping agents for  ${}^1\text{O}_2$ . From Fig. 9d, visible-light-irradiated  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  showed a boosted triplet signal which can be indexed to the signal of singlet oxygen ( ${}^1\text{O}_2$ ) generation as compared with pristine  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi}_3\text{NbO}_7$  in dark [4]. As shown in Fig. 9e, the  $\text{DMPO} \cdot \text{OH}$  confirmed the 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  exhibited the forceful intensities as

compared to  $\text{Bi}_3\text{NbO}_7$ , which reveals that more  $\cdot\text{OH}$  were produced. The ESR spectra of  $\text{O}_2^-$  and  ${}^1\text{O}_2$  are in well consistent with the results of chemical trapping experiments. From the above discussion, it can be summarized that 0D/2D  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites could generate more  $\text{O}_2^-$  and  ${}^1\text{O}_2$  produced by molecular oxygen. For further understanding the molecular oxygen activation, NBT was chosen to quantify the  $\text{O}_2^-$  concentration generated by  $\text{Bi}_3\text{NbO}_7$  and  $\text{Bi}/\text{Bi}_3\text{NbO}_7$





**Fig. 10.** (a) Photoluminescence spectroscopy of  $\text{Bi}_3\text{NbO}_7$  and X%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  ( $X = 1, 4, 7$  and  $10$ ); (b) Time-resolved transient PL decay spectroscopy of  $\text{Bi}_3\text{NbO}_7$  and 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$ .

composite [4]. The concentration of  $\cdot\text{O}_2^-$  produced with irradiation time can be determined, as shown in Fig. 9f. As expected, the  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  possesses a much higher  $\cdot\text{O}_2^-$  concentration than that of  $\text{Bi}_3\text{NbO}_7$  and blank one, which can be induced by its unique 0D/2D structure and faster interfacial charge transfer. It can be well confirmed that the  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites boosted the activation of molecular oxygen into  $\cdot\text{O}_2^-$  species.

Photoluminescence emission spectroscopy (PL) and time-resolved transient PL decay spectra of samples were exhibited in Fig. 10 [44–47]. The relative weak PL intensity indicates a low recombination rate of the photoinduced carriers. Distinctly, the pure  $\text{Bi}_3\text{NbO}_7$  and other X%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites both show PL emission peak at 470 nm (compliance with the band gap of  $\text{Bi}_3\text{NbO}_7$  from UV–vis diffuse reflectance results), but  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  samples display lower intensity than that of  $\text{Bi}_3\text{NbO}_7$  (Fig. 10a). This phenomenon indicates that the recombination rate of photogenerated carriers was efficiently restrained through constructing the 0D/2D  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites, which is consistent with the result of photocatalytic degradation test. Photogenerated carriers' separation process was also studied by the time-resolved transient PL spectroscopy. The emission lifetimes of 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  are shorter than the corresponding lifetimes of  $\text{Bi}_3\text{NbO}_7$  nanosheets (short lifetime  $\tau_1 = 1.94$  ns and long lifetime  $\tau_2 = 7.63$  ns for 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  versus short lifetime  $\tau_1 = 2.82$  ns and long lifetime  $\tau_2 = 8.51$  ns for  $\text{Bi}_3\text{NbO}_7$ ). It is obvious that the  $\tau$  value of  $\text{Bi}_3\text{NbO}_7$  nanosheets is decreased from 7.34 ns to 6.71 ns after introducing the Bi nanodot. This observed shortening of emission lifetime reveals the emergence of the electronic interaction between 2D  $\text{Bi}_3\text{NbO}_7$  nanosheets and 0D Bi nanodots [43]. What's more, the apparent electron transfer rate ( $k_{\text{ET}}$ ) in 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composites can be calculated according to the equation:

$$k_{\text{ET}} = \frac{1}{\tau(7\%\text{Bi}/\text{Bi}_3\text{NbO}_7)} - \frac{1}{\tau(\text{Bi}_3\text{NbO}_7)}$$

The apparent electron transfer rate value is  $1.61 \times 10^7 \text{ s}^{-1}$ , illustrating that the formed interface between 0D Bi nanodots and 2D  $\text{Bi}_3\text{NbO}_7$  nanosheets is beneficial to the effective electron transfer quenching of the excitation state of  $\text{Bi}_3\text{NbO}_7$  [20]. This fast electron-injection progress confirms that the 0D/2D structure remarkably accelerates charge transfer efficiency, thereby making for the molecular oxygen activation [46]. Both the lower PL intensity and the decreased fluorescence lifetime of  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  illustrated that the photogenerated carriers have been efficiently separated, which are superior to those of bare  $\text{Bi}_3\text{NbO}_7$ .

The transient photocurrent and electrochemical impedance spectroscopy (EIS) of the  $\text{Bi}_3\text{NbO}_7$  and 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composite were exhibited in Fig. 11. The photocurrent density of 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  (Fig. 11a) composite was much higher than those of  $\text{Bi}_3\text{NbO}_7$ , which

indicates that the  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  could lead to generating more charge carriers [48]. Generally, a smaller Nyquist curve indicated a flatter charge transfer resistance, which can be employed to study the dynamics of the localization state charges in the interface of the semiconductors. As shown in Fig. 11b, the sample 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  possesses a smaller arc radius than  $\text{Bi}_3\text{NbO}_7$ , which reveals that the introduction of 0D Bi nanodots could shorten the recombination rate of photo-induced electron-hole pairs and dramatically enhance the photoelectrochemical performance of  $\text{Bi}/\text{Bi}_3\text{NbO}_7$ . Photoelectrochemical tests are in well consistent with the results of PL and time-resolved transient PL decay tests.

Based on the experimental results and the theory calculations above, a reasonable mechanism was presented in Scheme 2. The valance band spectrum of pure  $\text{Bi}_3\text{NbO}_7$  was exhibited in Fig. S7. The electrons are excited from the valance band (VB) to the conduction band (CB) of  $\text{Bi}_3\text{NbO}_7$  under visible light excitation, leaving the holes on the VB of  $\text{Bi}_3\text{NbO}_7$  [21]. For pure  $\text{Bi}_3\text{NbO}_7$ , the conduction band were calculated to be  $-0.49$  eV, which is much lower than the Fermi level of semimetal Bi nanodots ( $-0.17$  eV). This relative potential refers to that the free electrons can easily transfer from  $\text{Bi}_3\text{NbO}_7$  to semimetal Bi, which coincides with the previous study on interfacial electron transfer from the surface of semiconductor to the plasmonic metal [22,26]. Furthermore, the semimetal Bi nanodots can plays as a charge sink for continuously adopting the electrons induced from the CB of the  $\text{Bi}_3\text{NbO}_7$  nanosheet. Meanwhile, the adsorbed molecule oxygen can trap these electrons to reduce to superoxide radical, and the holes left on the VB of  $\text{Bi}_3\text{NbO}_7$  reacted with hydroxyl groups to generate hydroxyl radical, which are both crucial oxidants for photocatalytic antibiotics degradation [33,49]. Furthermore, sustainable singlet oxygen ( $^1\text{O}_2$ ) generation could make by the process of charge transfer (that is, the direct oxidation of  $\cdot\text{O}_2^-$  by holes), which means energy transfer is also a sustainable way for  $^1\text{O}_2$  generation [42,49]. However, the photocatalytic activity of  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  is hampered when the semimetal Bi nanodots content is further increased. This phenomenon demonstrates that the excessive semimetal Bi nanoparticles on the surface of  $\text{Bi}_3\text{NbO}_7$  nanostructures can block the active sites of  $\text{Bi}_3\text{NbO}_7$  to generate carriers and this phenomenon has also been observed in other semiconductor/plasmonic metal system [50–52]. Meanwhile, the high surface area of photocatalysts can afford more adsorption sites for target pollutants [53]. Fig. S8 shows nitrogen adsorption-desorption isotherms of the pure  $\text{Bi}_3\text{NbO}_7$  and 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  sample. The specific surface areas of pure  $\text{Bi}_3\text{NbO}_7$  and 7%  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  are 28.32 and 39.76  $\text{m}^2/\text{g}$ , respectively. It can be observed that the higher specific surface areas in photocatalytic system also play a crucial part in the enhanced photocatalytic performance of the  $\text{Bi}/\text{Bi}_3\text{NbO}_7$  composite. The formed 0D semimetal Bi nanodots/2D  $\text{Bi}_3\text{NbO}_7$  nanosheets heterojunctions highly

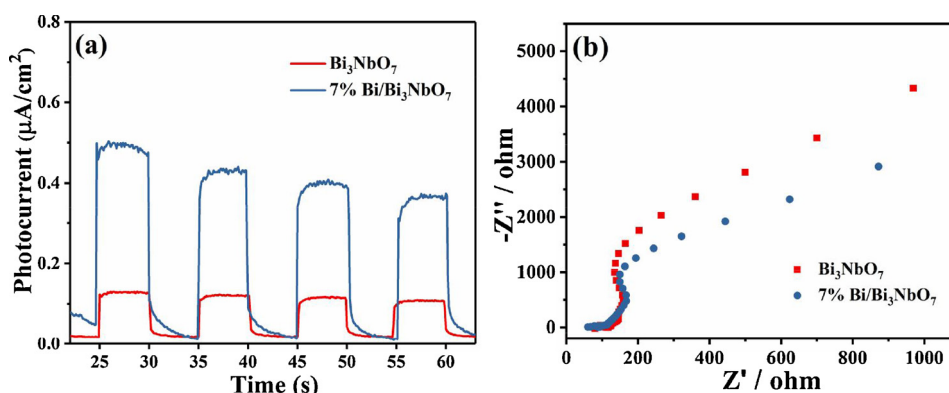
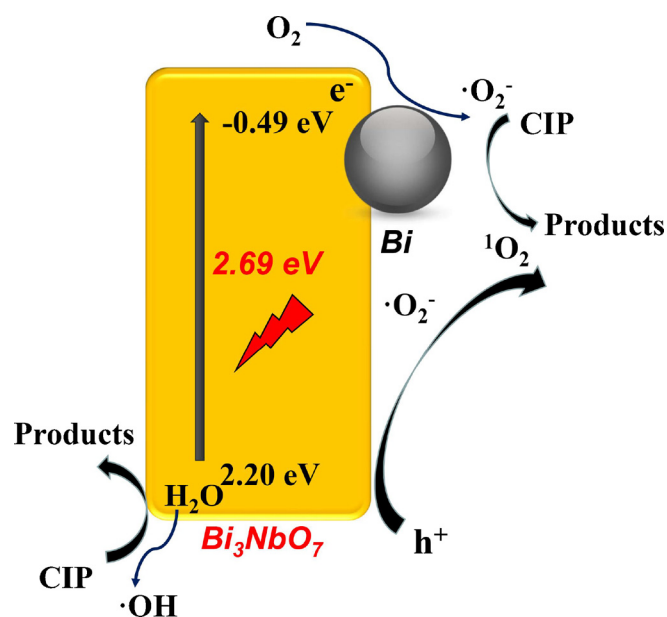


Fig. 11. (a) Transient photocurrent response and (b) Electrochemical impedance spectra of the Bi<sub>3</sub>NbO<sub>7</sub> and 7% Bi/Bi<sub>3</sub>NbO<sub>7</sub> under visible-light irradiation.



Scheme 2. Photocatalytic mechanism for Bi/Bi<sub>3</sub>NbO<sub>7</sub> composite under visible light irradiation.

enhance the carriers' separation efficiency and further boost the molecular oxygen activation of the Bi/Bi<sub>3</sub>NbO<sub>7</sub> composite [54]

#### 4. Conclusions

A series of 0D/2D heterojunctions of Bi nanodot/Bi<sub>3</sub>NbO<sub>7</sub> nanosheet are successfully synthesized through a simple two-step wet chemical method. The semimetal Bi nanodots are distributed uniformly on the surface of Bi<sub>3</sub>NbO<sub>7</sub> nanosheets. A strong covalent interaction exists between the Bi atom of Bi nanodots and Bi–O layer on the surface of the Bi<sub>3</sub>NbO<sub>7</sub> nanosheets, forming a charge transfer channel to transfer from Bi<sub>3</sub>NbO<sub>7</sub> to metal Bi nanodots. The introduction of semimetal Bi increased the visible light absorption of the photocatalysts and significantly promoted molecular oxygen activation of Bi<sub>3</sub>NbO<sub>7</sub> and improved the photocatalytic performance for degradation of CIP. The 0D/2D composite showed superior ·O<sub>2</sub><sup>-</sup> and <sup>1</sup>O<sub>2</sub> generation as compared with pristine Bi<sub>3</sub>NbO<sub>7</sub> under visible light irradiation. The photocatalytic degradation rate of ciprofloxacin (CIP) over the Bi/Bi<sub>3</sub>NbO<sub>7</sub> composites is 4.58 times higher than that over the pristine Bi<sub>3</sub>NbO<sub>7</sub>. The composite still exhibited high stability and photocatalytic activity even after five cycles. This work not only understands the mechanism of the molecular oxygen activation in metal-semiconductor system, but also paves a way to design highly efficient 0D/2D photocatalysts for environmental remediation.

#### Acknowledgements

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#### Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi: <https://doi.org/10.1016/j.apcatb.2018.08.063>.

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